



Detection of defects and deuterium in displacement-damaged tungsten by applying Rutherford backscattering spectroscopy and nuclear reaction analysis in channeling configuration

DeHydroC
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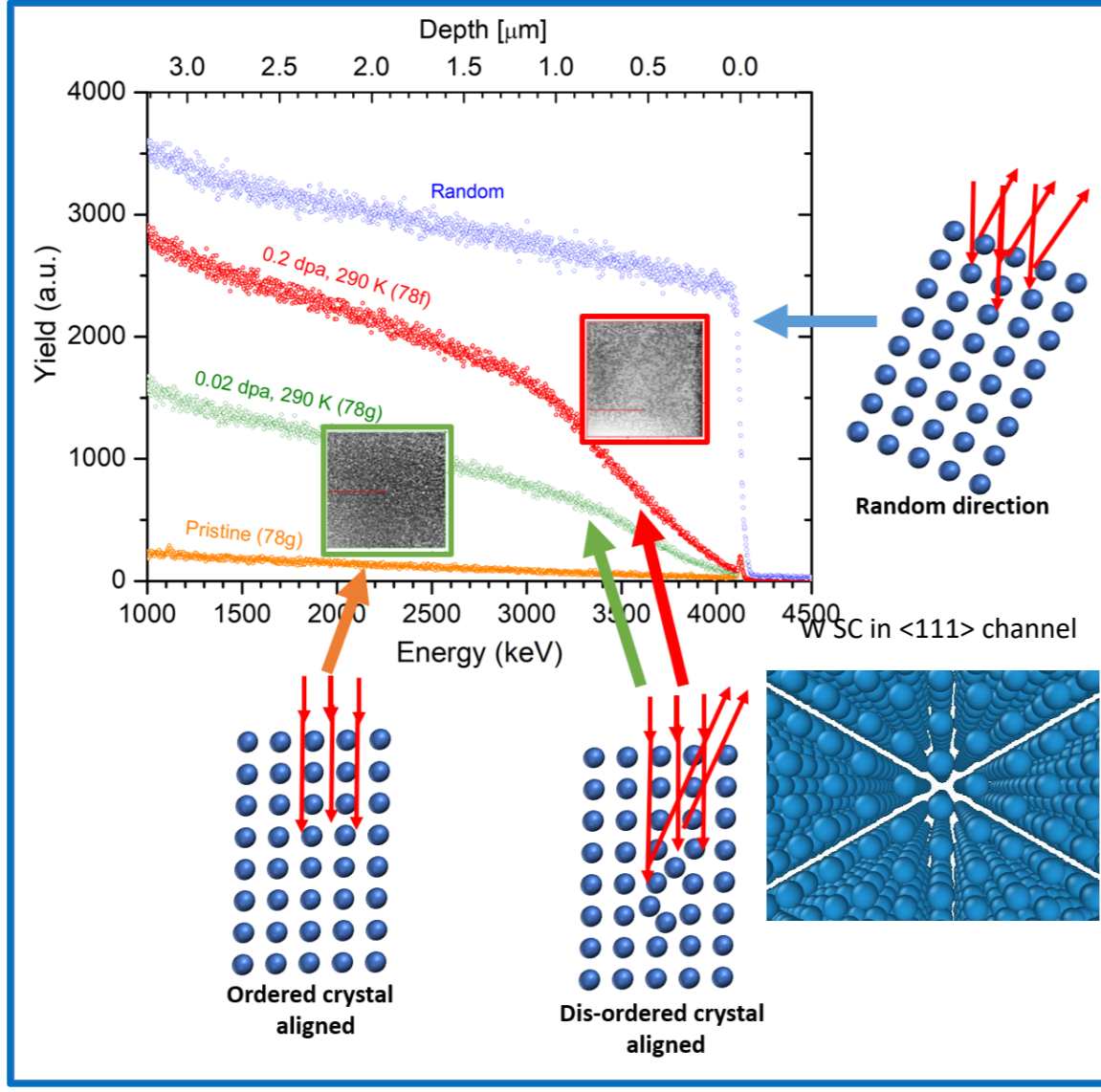
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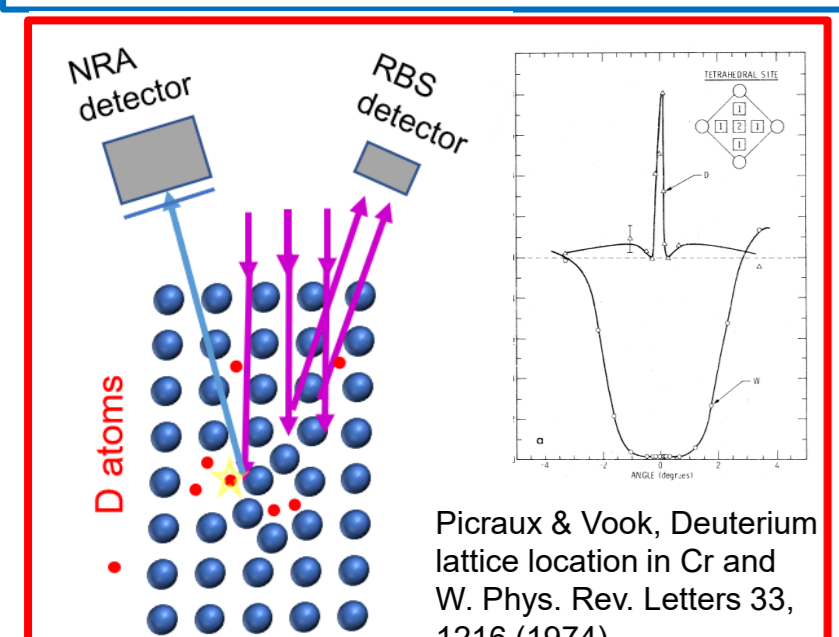
Introduction

In a future nuclear environment, 14 MeV neutrons from the D-T fusion reaction will create defects in the W lattice, affecting the physical properties of the material. Understanding the interaction of hydrogen with the host lattice is crucial for fusion research, since low hydrogen isotope (HI) retention is a stringent requirement for a fusion reactor. In this work, we show first results of the development of an advanced characterisation techniques to study lattice disorder and HI location. By combining different ion beam techniques in the channeling configuration, the influence of structural defects on HI retention and vice versa is studied.

To examine the generated defects created by high energy W ion irradiation in the W lattice, we utilized Rutherford backscattering spectrometry in channeling configuration (RBS-C), a well-established method for studying lattice disorder and defect evolution induced by irradiation. To quantify the disorder, the change in the ion yield of light ions backscattered along a specific crystallographic direction is measured [1].



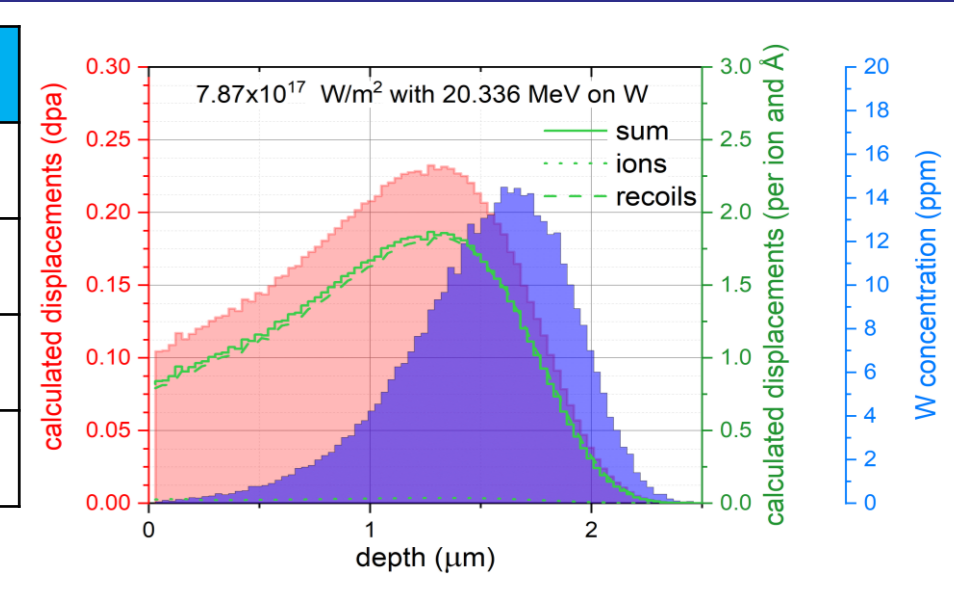
The use of nuclear reaction analysis (NRA-C) in conjunction with the RBS-C can provide information on the location of other species in the host material. In our case, we employed the ³He nuclear reaction with deuterium to study the position of deuterium in the tungsten lattice.



Sample production and irradiation

- W (111) crystals (Surface Preparation Laboratory B.V.) and W (100) crystals (MaTeck) were vibrato-polished and annealed at 2350 K for 5 min in UHV.
- Defects were created by irradiation with 10.8 MeV W³⁺ ions. Sample was tilted by 7° and rotated by 11° to minimize channeling.

Sample	Predominant defect expected
78f	"heavily damaged standard": 0.2 dpa, 290K
78g	"single vacancies": 0.02 dpa, 290 K
78c	"small vacancy clusters": 0.02 dpa, 800 K
78b	"big vacancy clusters": 0.2 dpa, 800 K



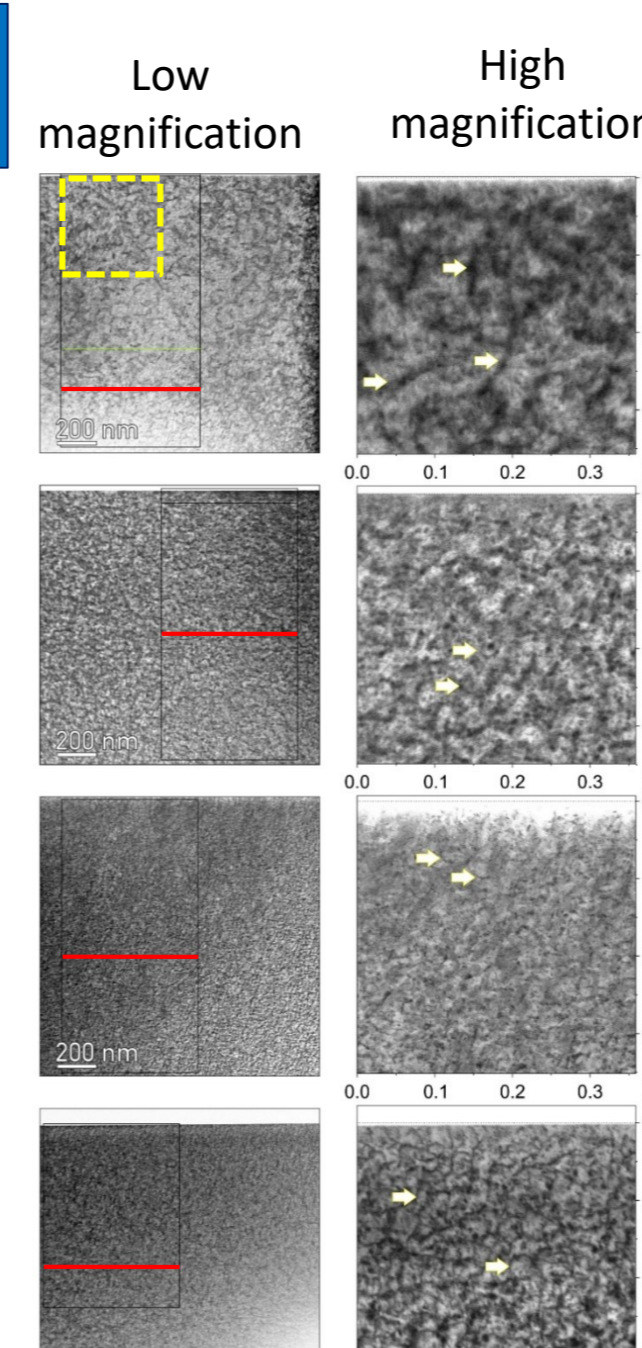
TEM ANALYSIS

78f (0.2 dpa, 290 K)
Depth ≈ 1.1 μm

78g (0.02 dpa, 290 K)
Depth ≈ 0.7 μm

78c (0.02 dpa, 800 K)
Depth ≈ 0.8 μm

78b (0.2 dpa, 800K)
Depth ≈ 1.1 μm

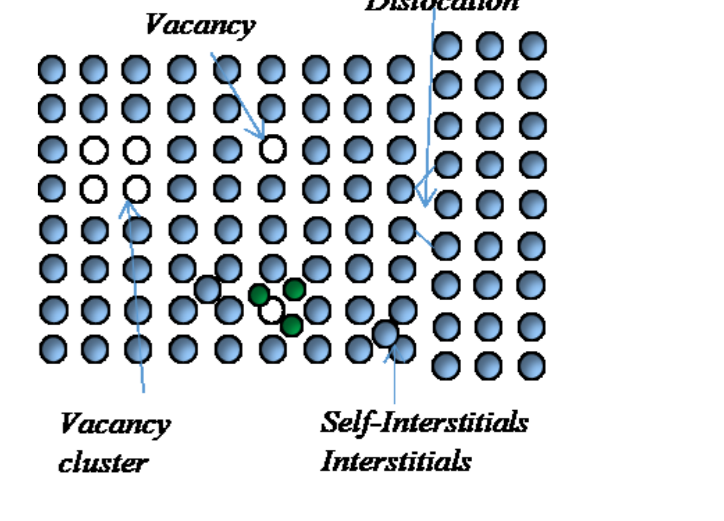


Dense network of dislocation lines (DL) (~100 nm), which are already nicely visible at low-mag overview micrographs. DL directions coincide with 110 >111 > 111-2 > 111 planes.

Only very short dislocation lines (~20-30 nm) in <111>, with prevailing U-shaped loops around "black dots" (size ~10 nm)

Dislocations: mainly dots and several isolated lines (< 50 nm, in <111> direction). Dots are smaller, several nm.

Several dislocation lines and larger black dots. DL lines are forming a network in <111>, forming square "polygons" with ~30 nm edge.

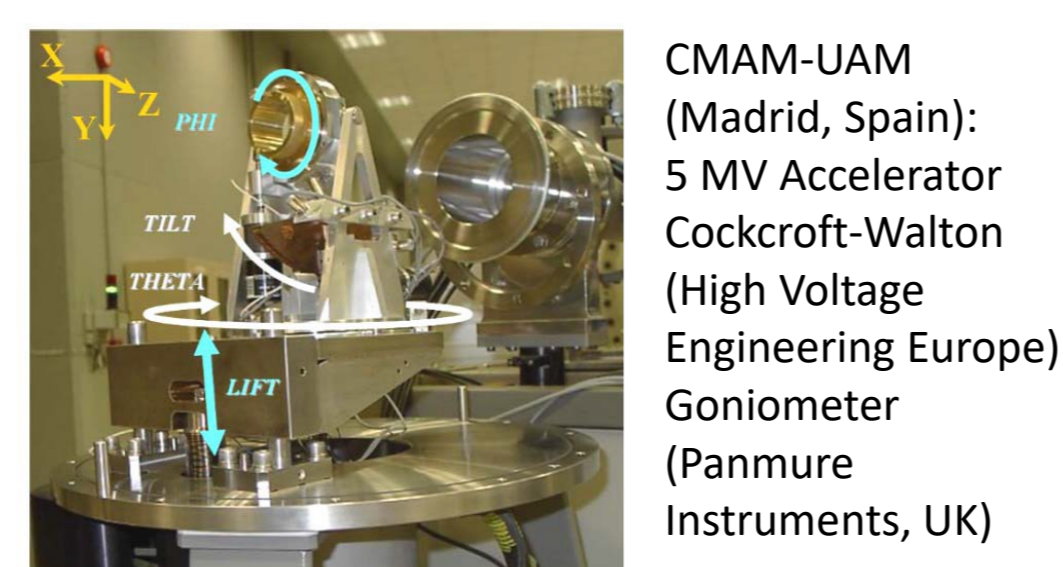


Channeling Rutherford Backscattering Spectroscopy (RBS-C)

Results

Channeling Nuclear Reaction Analysis (NRA-C)

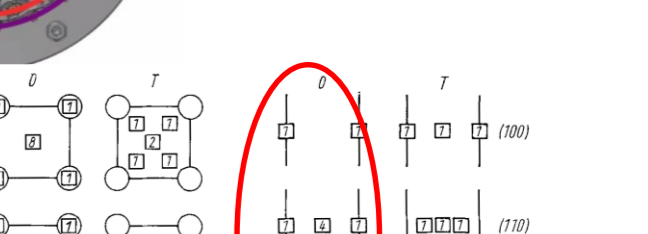
- Multi-energy RBS-C analysis along the <111> direction
- Four He⁺⁺ beam energies of 4.5, 4.0, 3.5, and 3.0 MeV
- W (111) single crystals: W-irradiated.
- The response of the induced structural damage signal versus analysing energy gives important information about the extension of the defects (uncorrelated or extended defects) [1,4].



- Simultaneous RBS-C and NRA-C measured at the Hedgehog setup for RBS channeling at ion beam center at HZDR.
- Samples: W (100) single crystal: W-irradiated and exposed to 5 eV/D plasma at 370 K

Multi Detector RBS set-up @ HZDR

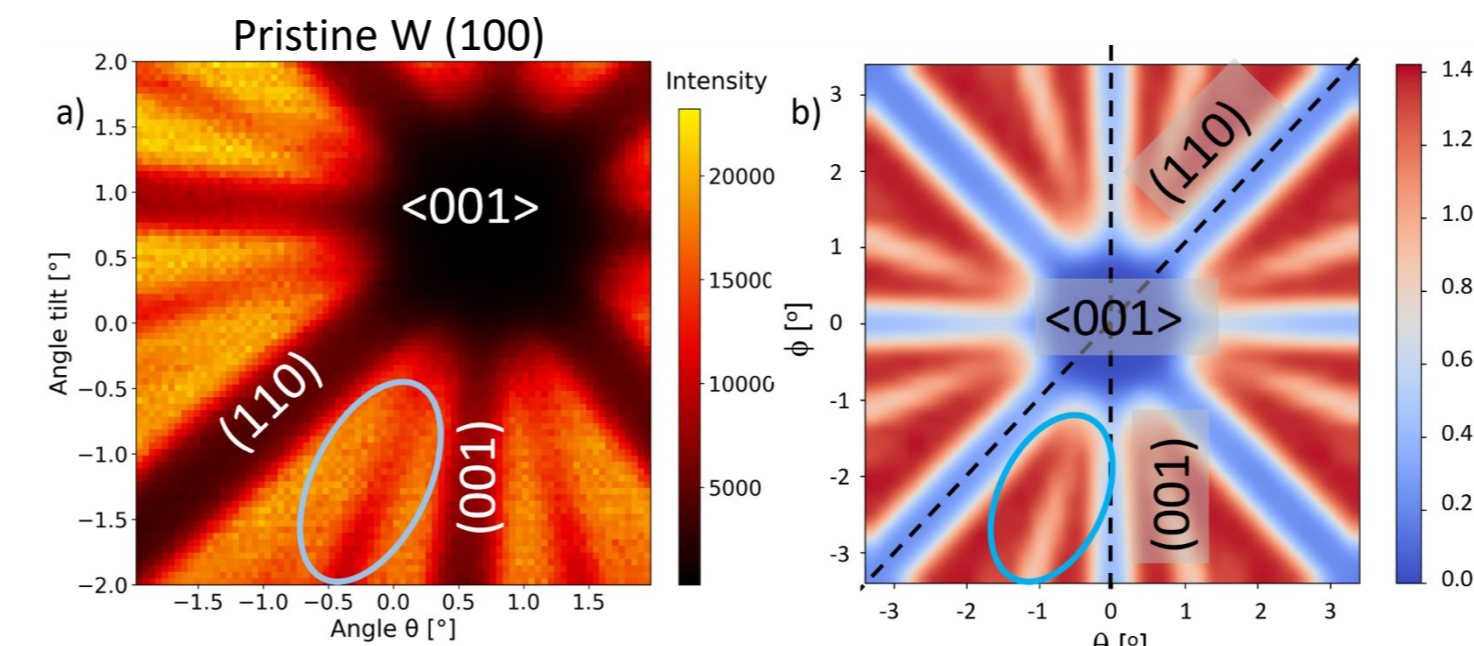
6 x 166° = 53.8 msr
12 x 150° = 107.6 msr
16 x 135° = 143.5 msr
20 x 120° = 179.3 msr
22 x 105° = 197.3 msr



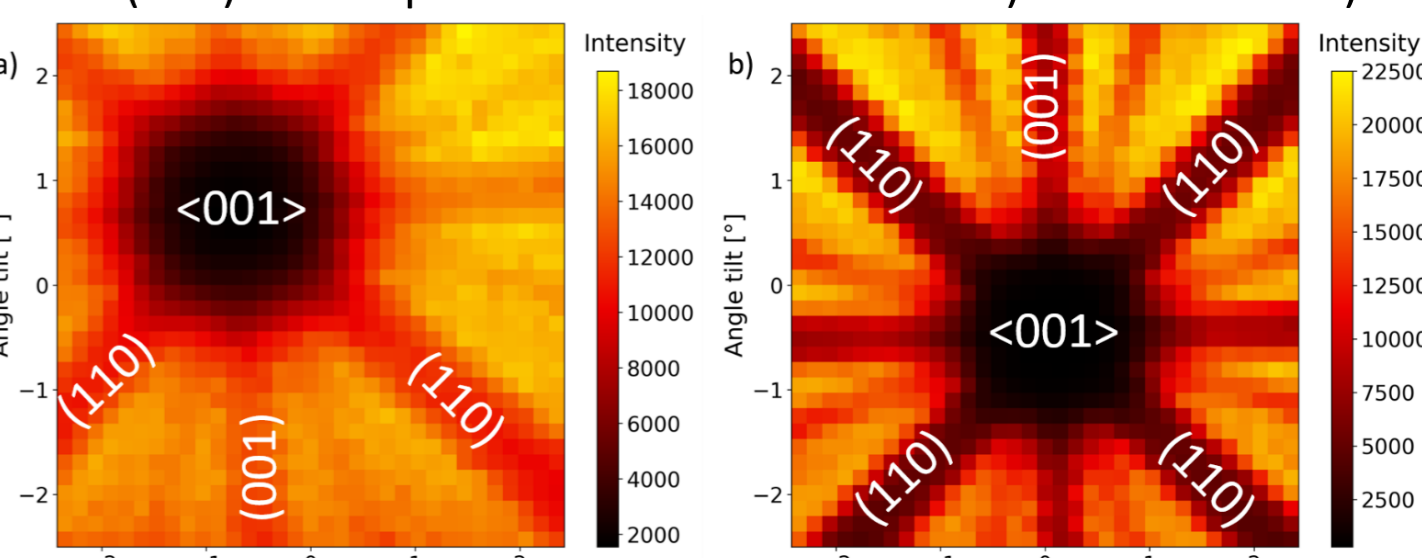
- W (100) single crystal should show a difference in the C-NRA signal when D is near the octahedral (OIS) or tetrahedral interstitial sites (TIS) [3].

⁴He 2.5 MeV RBS-C 2D maps / search for the channel

Measured (a) and simulated (b) RBS-C 2D maps of the RBS yield.



W (100) SC samples W-irradiated at 290 K a) and at 800 K b)

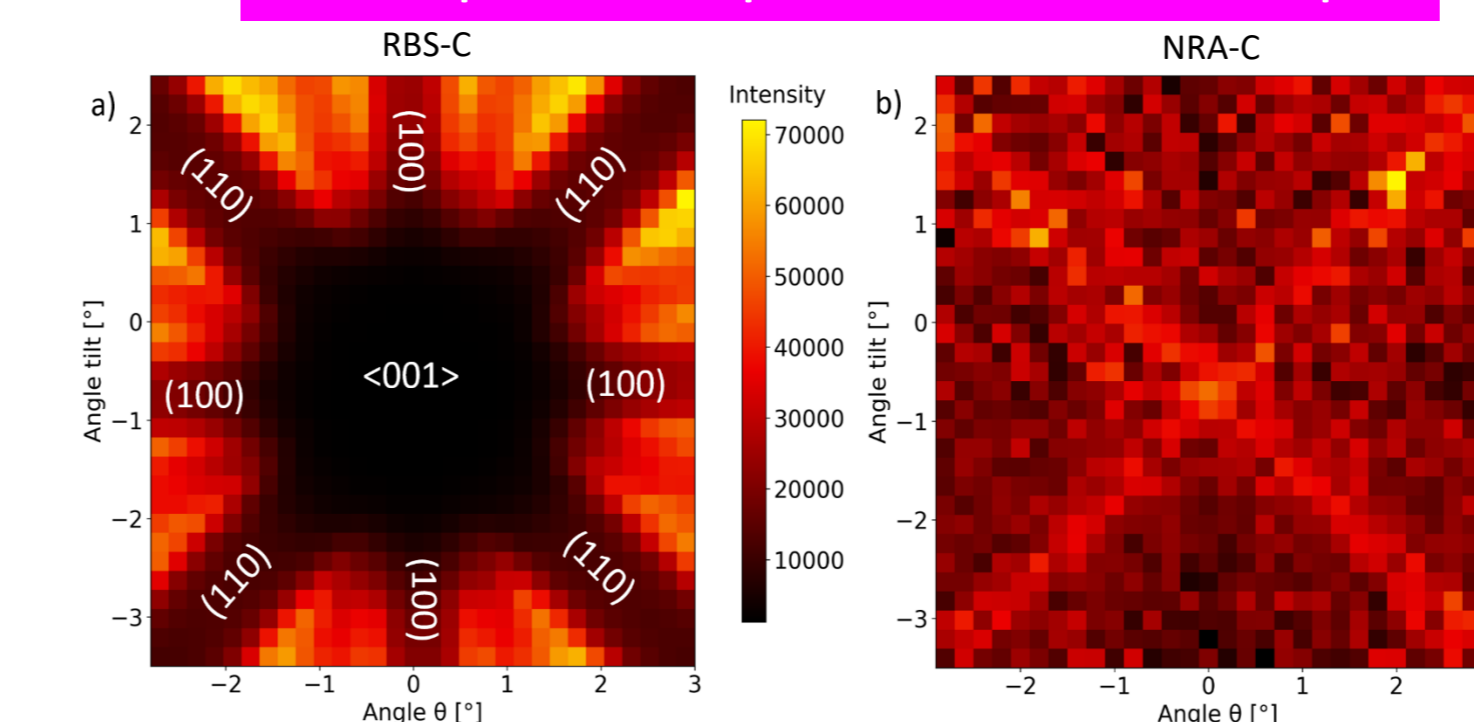


Less pronounced planar channels for damaged sample

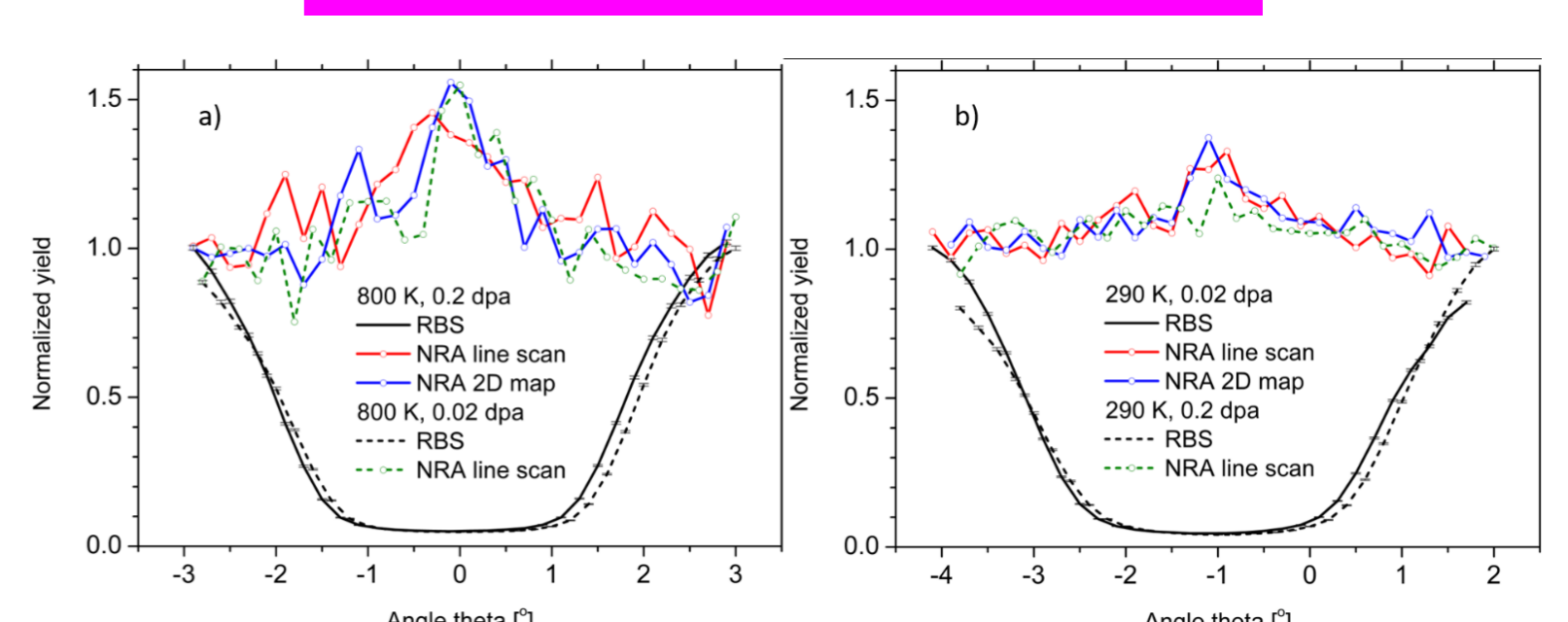
³He 0.8 MeV – simultaneous RBS-C and NRA-C 2D maps

Nuclear reaction D(³He,p)⁴He was used for this purpose

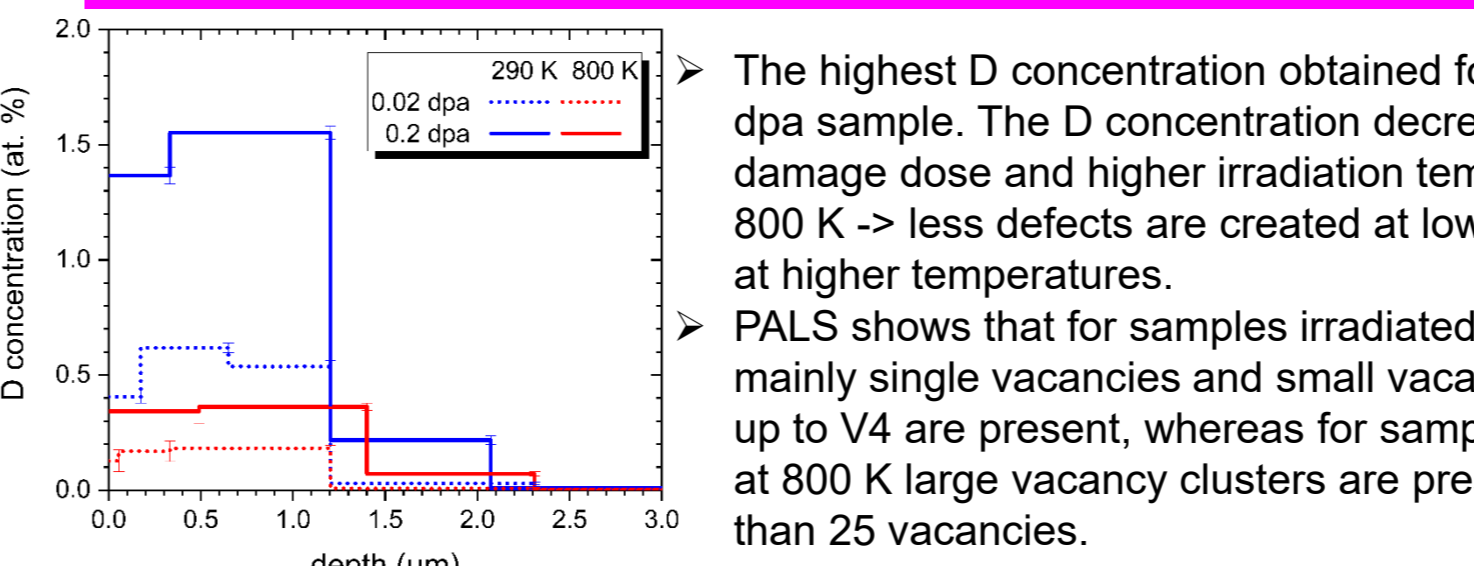
2D map for sample #2: 800 K, 0.2 dpa



Line scans of RBS-C and NRA-C



D depth profiles – classical ³He measurements

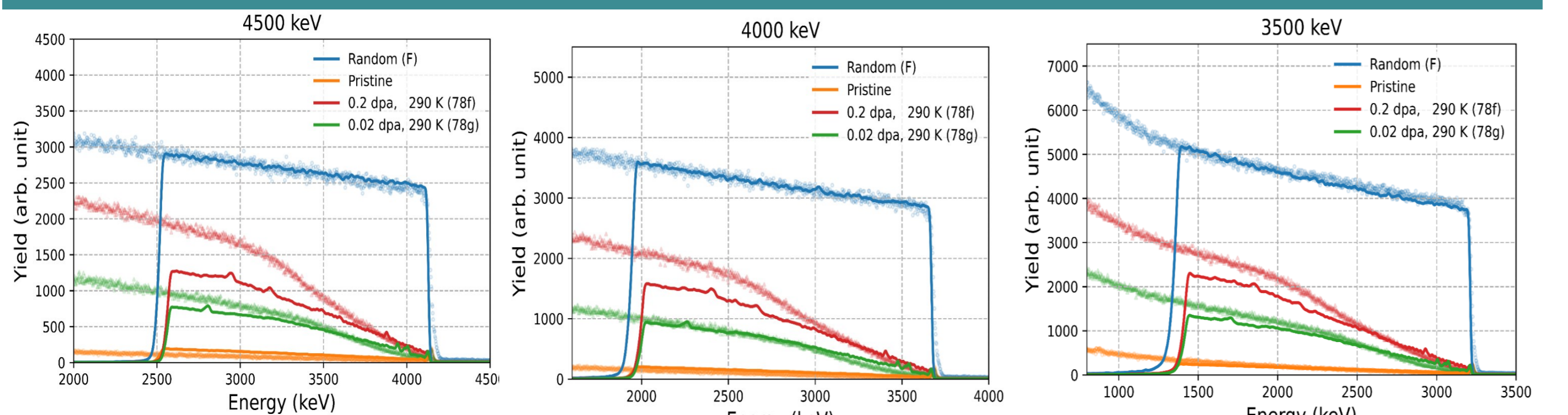


Angular scans across the <100> axial channel:

- 800 K damaged samples: NRA yield peaks in the centre of the channel, where the RBS is at its minimum.
- Room-temperature-irradiated samples: NRA yield peaks in the centre of the channel for the low dose. For the high dose sample, hardly any peak is observed.

For more details see [6]

RBSADEC simulation with Molecular Dynamics (MD) simulations as input



RBSADEC code [4] results (lines) compared with ⁴He RBS/C spectra (circles). As input, realistic damage structures produced with overlapping molecular dynamic cascade simulations (MD) were used [Granberg et al. J. Nucl. Mater. 556 (2021) 153158].

For more details see [5]

Conclusions

Radiation-induced defect production in W was studied by a combination of experimental and simulation methods. The analysis of structural defects was performed using multi-energy Rutherford backscattering spectroscopy in channeling configuration (multi-energy C-RBS). Detailed transmission electron microscopy (TEM) analysis of the samples revealed the presence of dislocation lines and loops of different sizes. The RBSADEC code was used to simulate the measured C-RBS spectra. For the first time for W, molecular dynamics (MD) simulations of overlapping cascades were used as input. With MD, a very good agreement between the simulated and experimental spectra was obtained for the sample prepared at a lower dose. A discrepancy is observed for the high-dose-irradiated sample, which is ascribed to the presence of extended defects such as dislocation lines, which are clearly observed by TEM, but cannot be formed in finite size MD cells [5].

NRA-C using a ³He probing beam was measured for displacement-damaged W where the created defects were before decorated with D in order to determine the position of D at defects. Maximum signal was obtained in the <100> axial channel and in the (110) planar channel. RBSADEC code was upgraded in order to model the NRA-C spectra [Jin et al. Phys. Rev. Mater. 2024]. RBSADEC simulations in combination with density functional theory are in progress to study the position of hydrogen in vacancies and vacancy clusters is in progress. A first comparison indicates that D is positioned close to tetrahedral sites.

References

[1] Feldman et al., Academic Press, San Diego, (1982), pp. 88–116.

[2] Hu et al., J. Nucl. Mater. 556, (2021) 153175.

[3] Cabstanjen, Phys. Stat. Sol. 59, (1980) 11–26.

[4] Zhang et al., Phys. Rev. E 94, (2016) 043319.

[5] Markelj et al., Acta Materialia 263 (2024) 119499.

[6] Markelj et al., Nucl. Mater. Energ. 39 (2024) 101630.



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